

# Superhard $sp^2$ - $sp^3$ hybridized $BC_2N$ : A 3D crystal with 1D and 2D alternate metallicity

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A novel  $sp^2$ - $sp^3$  hybridized orthorhombic  $BC_2N$  ( $o$ - $BC_2N$ ) structure (space group:  $Pmm2$ , No. 25) is investigated using first-principles calculations.  $o$ - $BC_2N$  is constructed from multi-layers of C sandwiched between two layers of BN along the  $c$  axis; this structure contains  $sp^2$ - and  $sp^3$ -hybridized B-C, C-C, and C-N bonds. The structural stability of  $o$ - $BC_2N$  is confirmed based on the calculation results for elastic constants and phonon dispersions. On the basis of the semi-empirical microscopic model, we speculate that the  $o$ - $BC_2N$  compound is a potential superhard material with a Vickers hardness of 41.2 GPa. Calculated results for electronic band structures, density of states (DOS) and partial DOS (PDOS) show that the  $o$ - $BC_2N$  crystal is metallic. The conducting electrons at the Fermi level are mostly from the  $2p$  orbits of  $sp^2$ -hybridized  $B_4$ ,  $N_1$ , and  $C_i$  ( $i = 2, 3, 4, 6, 7, 8$ ) atoms, with slight contribution from the  $sp^3$ -hybridized  $B_2$  atoms. Furthermore, the calculated electron orbits of the  $o$ - $BC_2N$  crystal demonstrate that the  $2p$  orbits of the  $sp^2$ -hybridized atoms overlapped and formed  $\pi$  bonds. The electrons can conduct through the  $\pi$  bonds along the orientation parallel to the [100] and [010] directions in different layers, and the basal planes were formed by  $B_2$ - $C_3$ - $C_4$  blocks, indicating that the  $o$ - $BC_2N$  possesses the fascinating electronic property of linear-planar metallicity. Published by AIP Publishing. [<http://dx.doi.org/10.1063/1.4984760>]

## I. INTRODUCTION

Ternary boron-carbon-nitrogen (B-C-N) compounds, which are anticipated to possess extreme hardness combined with adjustable band gap by changing their atomic composition and structure, have increasingly attracted attention due to their wide industrial application prospects.<sup>1–12</sup> Since the first synthesis of graphitic  $BC_2N$  ( $g$ - $BC_2N$ ) by Kouvetakis<sup>5,13</sup> and co-workers by using the chemical vapor deposition method, researchers have devoted efforts to obtain high-density  $BC_2N$  compounds experimentally<sup>14–16</sup> and theoretically.<sup>9,17–21</sup> Solozhenko *et al.*<sup>16</sup> have successfully synthesized  $c$ - $BC_2N$  crystals under high temperature and high pressure conditions by using  $g$ - $BC_2N$  as the raw material. The Vickers and Knoop hardness measurements suggested that the  $c$ - $BC_2N$  phase was harder than the  $c$ -BN single crystal but still softer than diamond. A possible hetero-diamond  $\beta$ - $BC_2N$  phase transformed from  $g$ - $BC_2N$  was predicted by Tateyama *et al.*<sup>19</sup> The  $\beta$ - $BC_2N$  phase displays a superior bulk modulus value of 438 GPa, which is close to that of diamond (464 GPa). Bond structure calculations showed that the  $\beta$ - $BC_2N$  displays a band gap of 3.97 eV. In addition, many  $BC_2N$  phases have been obtained by replacing C atoms with B and N atoms in a diamond structure. Sun *et al.*<sup>22</sup> have investigated several  $c$ - $BC_2N$  structures with a diamond-like unit cell by using the *ab initio* pseudopotential density functional method. Their results indicated that these  $c$ - $BC_2N$  phases were direct band-gap semiconductors or semimetals with superior Vickers

hardness values. Two novel types of  $o$ - $BC_2N$  crystals were predicted by Mattesini<sup>20</sup> and co-workers. Their results showed that these  $o$ - $BC_2N$  phases are possibly harder than  $c$ -BN, and their band gaps are 2.04 and 1.69 eV, respectively. The commonly proposed diamond-like  $BC_2N$  phases are either insulators or semiconductors due to electron localization in  $sp^3$  hybridized atoms and the isoelectronic properties of  $(BN)_2$  and C units.

Nevertheless, crystals with  $sp^2$ - $sp^3$  hybrid characteristics, such as T- $B_7N_7$ <sup>23</sup> and 3D-(n, 0) carbons,<sup>24</sup> exhibit metallicity because of the intensified electron delocalization of  $sp^2$  configuration with a short distance between the adjacent atomic layers. A novel orthorhombic  $B_3N_5$  ( $o$ - $B_3N_5$ ) structure (space group  $C222_1$ ) was recently reported by Li *et al.*<sup>25</sup> The  $o$ - $B_3N_5$  contains one type of B atoms and two types of N ( $N_1$  and  $N_2$ ) atoms. B and  $N_1$  atoms are four-coordinated, and each  $N_2$  atom is  $sp^2$  hybridized with two B atoms and one  $N_2$  atom. This  $sp^2$ - $sp^3$  hybridized structure possesses surprising properties, such as a narrow band gap of 0.775 eV, a high energy density of approximately 3.44 kJ/g, and an increased Vickers hardness of 44 GPa.

This paper reports on a novel  $sp^2$ - $sp^3$  mixed hybridization  $o$ - $BC_2N$  structure, which is produced by replacing a part of B and N atoms with C atoms in the  $o$ - $B_3N_5$  unit cell. The  $o$ - $BC_2N$  is a sandwich-like structure constructed from C layers alternately stacked with BN layers along the  $c$  axis.  $sp^2$ - and  $sp^3$ -hybridized B, C, and N atoms coexist in this structure. Using the first-principles density-functional method, we investigated the ground-state properties of the  $o$ - $BC_2N$  structure. The calculated results show that the  $o$ - $BC_2N$  structure displays

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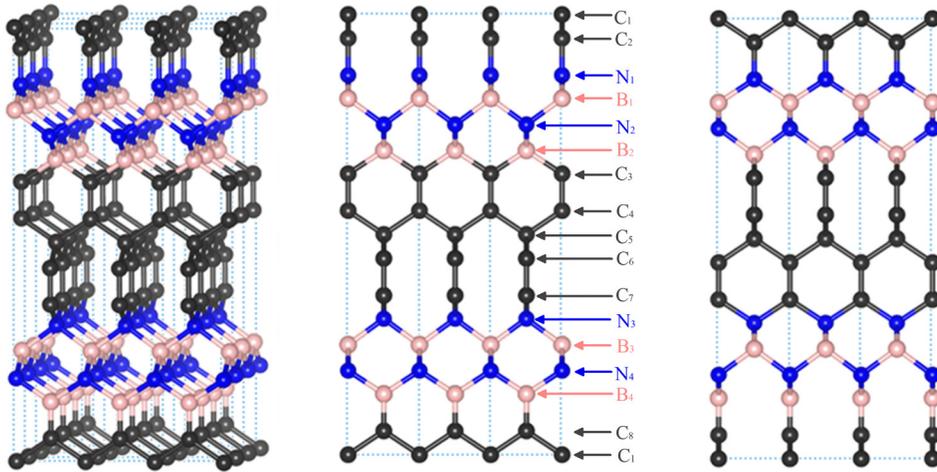


FIG. 1. The relaxed crystal structure (a) and views along the [010] (b) and [100] (c) directions of the *o*-BC<sub>2</sub>N. The boron, carbon, and nitrogen atoms are described as pink, black, and blue colors, respectively.

the distinctive electronic property of linear-planar metallicity. The Vickers hardness of the *o*-BC<sub>2</sub>N crystal is approximately 41.2 GPa, indicating that *o*-BC<sub>2</sub>N is a potential superhard metallic material.

## II. CALCULATION METHODS

First-principles calculations were performed using the pseudopotential density functional method<sup>26</sup> implemented in the CASTEP code.<sup>27</sup> The Perdew–Berke–Ernzerhof form of the generalized gradient approximation (GGA-PBE) was used to treat the exchange-correlation function.<sup>28</sup> To ensure an accurate determination of electronic properties, calculations are repeated using the hybrid Heyd–Scuseria–Ernzerhof functional (HSE06).<sup>29,30</sup> The ultrasoft pseudopotential<sup>31</sup> was expanded by the plane-wave basis set with a cutoff energy of 770 eV. Elastic constants, bulk modulus, and shear modulus can be calculated directly by using the CASTEP code. Structural relaxation was performed using the Broyden–Fletcher–Goldfarb–Shanno methods.<sup>32</sup> The phonon modes of the equilibrium crystal structure were calculated through finite displacement theory<sup>33</sup> using a  $k$ -point space ( $2\pi \times 0.015 \text{ \AA}^{-1}$ ) corresponding to the fine quality level. The coordinates of the high symmetry points in reciprocal space are  $Y(-0.5, 0, 0)$ ,  $G(0, 0, 0)$ ,  $Z(0, 0, 0.5)$ ,  $T(-0.5, 0, 0.5)$ ,  $R(-0.5, 0.5, 0.5)$ ,  $U(0, 0.5, 0.5)$ ,  $X(0, 0.5, 0)$ , and  $S(-0.5, 0.5, 0)$ . The supercell volume is 16 times that of the cell in the phonon calculations. The energy band structures and PDOS calculations used the  $k$  points which were selected with a  $9 \times 9 \times 2$  mesh in the Brillouin zone. The Vickers

hardness of the *o*-BC<sub>2</sub>N crystal is estimated using the microscopic theoretical hardness model.<sup>34–37</sup>

## III. RESULTS AND DISCUSSIONS

Figure 1 shows the *o*-BC<sub>2</sub>N crystal structure after structural relaxation, and Table I presents the atomic Wyckoff positions of *o*-BC<sub>2</sub>N. The *o*-BC<sub>2</sub>N structure exhibits a sandwich-like layer structure constructed from C layers separated from BN layers along the  $c$  axis. A number of  $sp^2$ - and  $sp^3$ -hybridized B, C, and N atoms formed a mixed hybridization *o*-BC<sub>2</sub>N structure. Table II shows the calculated lattice parameters, total energies  $E_t$ , bulk moduli  $B$ , and shear moduli  $G$  of *o*-BC<sub>2</sub>N, *o*-B<sub>3</sub>N<sub>5</sub>, diamond, and *c*-BN structures. For comparison, the experimental values for diamond and *c*-BN are also shown in Table II. The errors in lattice parameters, bulk moduli, and shear moduli between the calculated and experimental values for diamond and *c*-BN are very small, indicating that the calculated results are satisfactory. The calculated lattice constants  $a$ ,  $b$ , and  $c$  of *o*-BC<sub>2</sub>N are 2.627, 2.544, and 15.817 Å, respectively. The bulk modulus  $B$  of *o*-BC<sub>2</sub>N is 310 GPa, which is equal to that of *o*-B<sub>3</sub>N<sub>5</sub>. The shear modulus  $G$  of *o*-BC<sub>2</sub>N is 435 GPa lower than that of diamond but higher than that of *o*-B<sub>3</sub>N<sub>5</sub> and *c*-BN.

The formation energy ( $E_f$ ) of *o*-BC<sub>2</sub>N was calculated according to the formula  $E_f = E_{BC_2N} - (E_{c-BN} + 2E_C)$ , where  $E_{BC_2N}$ ,  $E_{c-BN}$ , and  $E_C$  represent the energy of BC<sub>2</sub>N, *c*-BN, and

TABLE I. Atomic Wyckoff positions of the *o*-BC<sub>2</sub>N structure.

Atom	Wyckoff positions	Atom	Wyckoff positions
B <sub>1</sub>	1a (0, 0, 0.807)	C <sub>1</sub>	1a (0, 0, -0.002)
B <sub>2</sub>	1d (0.5, 0.5, 0.691)	C <sub>2</sub>	1b (0, 0.5, 0.945)
B <sub>3</sub>	1b (0, 0.5, 0.251)	C <sub>3</sub>	1b (0, 0.5, 0.637)
B <sub>4</sub>	1c (0.5, 0, 0.141)	C <sub>4</sub>	1b (0, 0.5, 0.551)
N <sub>1</sub>	1b (0, 0.5, 0.862)	C <sub>5</sub>	1d (0.5, 0.5, 0.498)
N <sub>2</sub>	1c (0.5, 0, 0.752)	C <sub>6</sub>	1c (0.5, 0, 0.446)
N <sub>3</sub>	1d (0.5, 0.5, 0.310)	C <sub>7</sub>	1c (0.5, 0, 0.362)
N <sub>4</sub>	1a (0, 0, 0.195)	C <sub>8</sub>	1c (0.5, 0, 0.048)

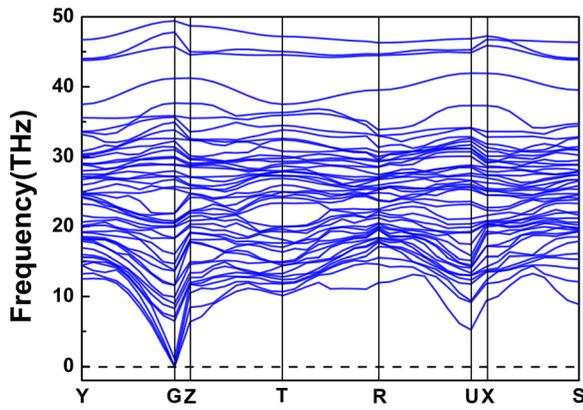
TABLE II. Lattice parameters, formation energy ( $E_f$ ), bulk modulus  $B$  and shear modulus  $G$  of the *o*-BC<sub>2</sub>N, *o*-B<sub>3</sub>N<sub>5</sub>, diamond and *c*-BN structures after structural optimization.

Structure	<i>o</i> -BC <sub>2</sub> N	<i>o</i> -B <sub>3</sub> N <sub>5</sub>	Diamond	<i>c</i> -BN
Symmetry	<i>Pmm</i> 2	<i>C</i> 222 <sub>1</sub>	<i>F</i> -43m	<i>F</i> -43m
$a$ (Å)	2.627	3.642 (3.638 <sup>a</sup> )	3.568 (3.567 <sup>b</sup> )	3.627 (3.615 <sup>b</sup> )
$b$ (Å)	2.554	3.639 (3.636 <sup>a</sup> )		
$c$ (Å)	15.817	15.883 (15.834 <sup>a</sup> )		
$E_f$ (eV/atom)	0.53	0.42		
$B$ (GPa)	310	309 (328 <sup>a</sup> )	424 (443 <sup>c</sup> )	369 (400 <sup>c</sup> )
$G$ (GPa)	435	266 (289 <sup>a</sup> )	522 (535 <sup>c</sup> )	380 (409 <sup>c</sup> )

<sup>a</sup>Reference 25.

<sup>b</sup>Reference 14.

<sup>c</sup>Reference 39.

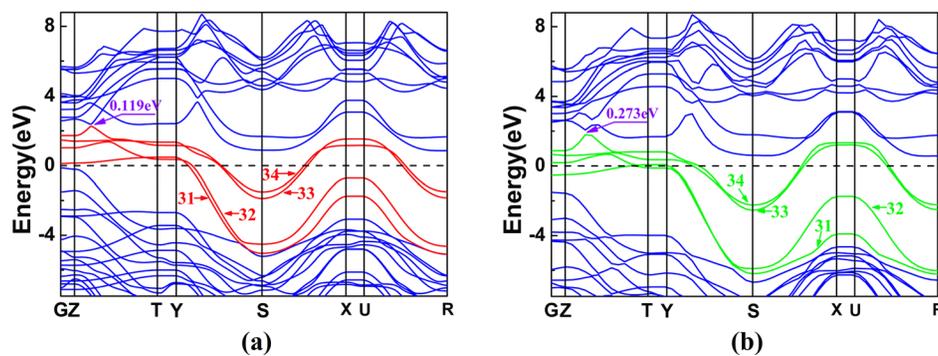
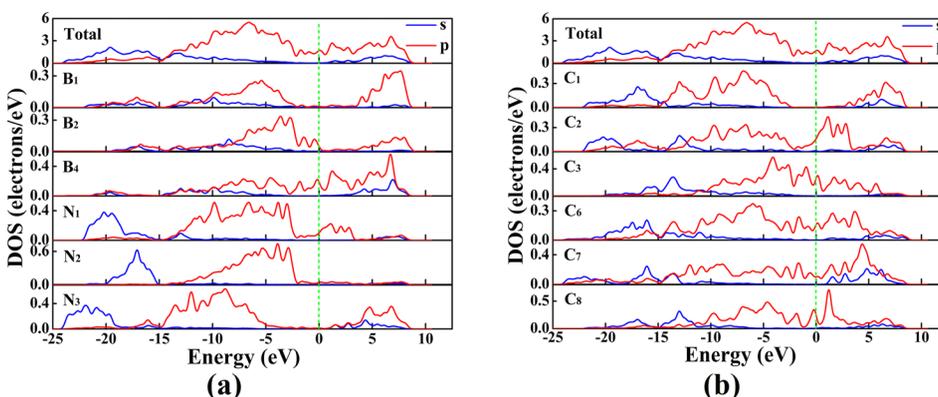
FIG. 2. The phonon dispersion curves for *o*-BC<sub>2</sub>N.

diamond formula unit, respectively. The positive formation energy indicates that the *o*-BC<sub>2</sub>N phase is metastable and tends to dissociate into *c*-BN and diamond. To check the structural stability of the *o*-BC<sub>2</sub>N phase, its elastic stiffness constants (GPa) are calculated as follows:  $C_{11} = 720.7$ ,  $C_{22} = 683.2$ ,  $C_{33} = 943.9$ ,  $C_{44} = 73.1$ ,  $C_{55} = 139.2$ ,  $C_{66} = 126.5$ ,  $C_{12} = 27.5$ ,  $C_{13} = 104.6$ , and  $C_{23} = 117.0$ . For the orthorhombic structures, the generalized elastic stability criteria is  $C_{ii} > 0$  ( $i = 1, 2, \dots, 6$ ),  $[C_{11} + C_{22} + C_{33} + 2(C_{12} + C_{13} + C_{23})] > 0$ ,  $(C_{11} + C_{22} - 2C_{12}) > 0$ ,  $(C_{11} + C_{33} - 2C_{13}) > 0$ , and  $(C_{22} + C_{33} - 2C_{23}) > 0$ .<sup>38</sup> The elastic constants of the *o*-BC<sub>2</sub>N structure satisfy the mechanical stability criteria, suggestive of its mechanical stability. The calculated phonon dispersion curves (Fig. 2) demonstrate that the *o*-BC<sub>2</sub>N structure is dynamically stable because of the absence of imaginary frequencies in the entire Brillouin zone.

To explore the electronic properties of the *o*-BC<sub>2</sub>N structure, the band structures are calculated by standard DFT and hybrid functional HSE06 methods. The GGA/PBE calculated results show that there are four occupied bands (marked 31, 32, 33, and 34) crossing the Fermi level in Fig. 3(a), which indicates that the *o*-BC<sub>2</sub>N crystal may possess metallicity. It is generally known that the standard DFT calculations would underestimate the band gap. Hence, we completed the verification calculation for the band structure of *o*-BC<sub>2</sub>N using the screened hybrid functional HSE06 method shown in Fig. 3(b). The result shows that the DFT calculation underestimated the band gap for 0.154 eV. Comparing both the band structures, we found no significant changes for the position of energy bands from 31 to 34, although the band gap value was underestimated by DFT calculation.

The calculated total density of states (DOS) and the partial DOS (PDOS) of B<sub>*i*</sub> ( $i = 1, 2, 4$ ), N<sub>*j*</sub> ( $j = 1 \sim 3$ ) and C<sub>*k*</sub> ( $k = 1 \sim 3, 6 \sim 8$ ) atoms in the *o*-BC<sub>2</sub>N structure are shown in Fig. 4. The PDOS of B<sub>3</sub>, N<sub>4</sub>, C<sub>4</sub>, and C<sub>5</sub> atoms are similar to those of B<sub>1</sub>, N<sub>3</sub>, C<sub>6</sub>, and C<sub>1</sub>, respectively. The conducting electrons at the Fermi level are mostly from the 2*p* electrons of B<sub>4</sub>, N<sub>1</sub>, and C<sub>*i*</sub> ( $i = 2, 3, 4, 6, 7, 8$ ) atoms with *sp*<sup>2</sup>-hybridized states. The *sp*<sup>3</sup>-hybridized B<sub>2</sub> atoms slightly contributed to the electron state at the Fermi level. The contribution of B<sub>1</sub>, B<sub>3</sub>, C<sub>1</sub>, C<sub>5</sub>, N<sub>2</sub>, N<sub>3</sub>, and N<sub>4</sub> is very small and can be negligible.

Figure 5 shows the electron orbits of four occupied bands crossing the Fermi level [31 to 34 in Fig. 3(a)] in the *o*-BC<sub>2</sub>N structure. The projection of the electron orbital along the *b* axis of the *o*-BC<sub>2</sub>N structure is shown in Fig. 5(a). We divided the projection into four regions marked

FIG. 3. The electronic structures of *o*-BC<sub>2</sub>N: GGA/PBE (a) and HSE06 (b). The Fermi level is indicated by the horizontal dashed line.FIG. 4. Total density of states, *s* (blue line) and *p* (red line) partial density of states of the *o*-BC<sub>2</sub>N structure. The Fermi level is indicated by the green dashed line.

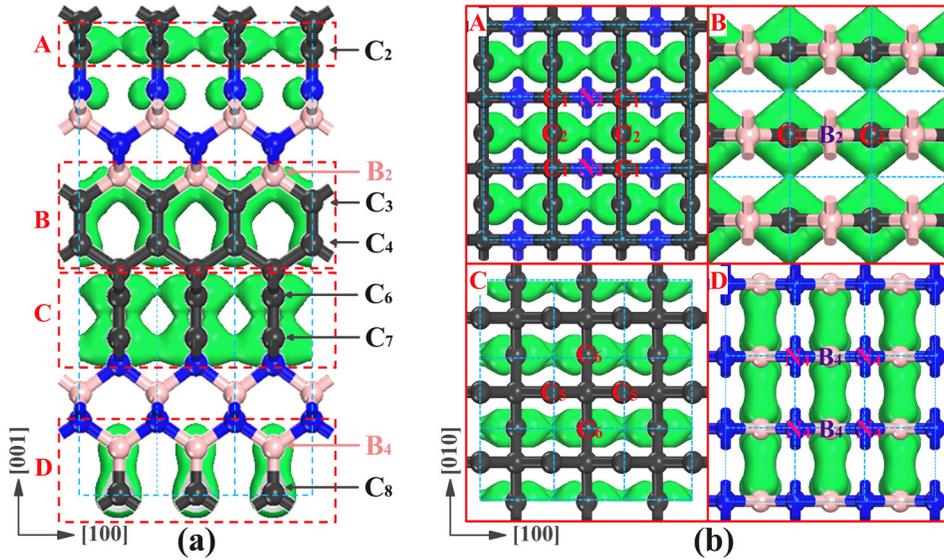


FIG. 5. Calculated electron orbitals of the *o*-BC<sub>2</sub>N structure. Panel (a) shows the projection of an electron orbital viewing along the [010] direction and panel (b) shows four projections of the electron orbitals viewing along the [001] direction which corresponds to the local fragments [marked red rectangle squares in panel 5(a)].

with A, B, C, and D. In the A, B, and C regions, the electron orbits are overlapped along the [100] crystal orientation and envelop the C<sub>2</sub> atom chain, the B<sub>2</sub>-C<sub>3</sub>-C<sub>4</sub> triatomic layer and the C<sub>6</sub>-C<sub>7</sub> diatomic layer, respectively. In the D region, the electron orbit only covers the B<sub>4</sub>-C<sub>8</sub> diatomic units. It is worth noting that the electron orbits crossing the Fermi level in the regions of A, B, C, and D are disconnected in the [001] crystal orientation of the *o*-BC<sub>2</sub>N structure, which implies that the conduction electrons are blocked in the *c* axis of the *o*-BC<sub>2</sub>N crystal. The electron orbits in A, B, C, and D areas are projected along the *c* axis of the crystal structure and presented in Fig. 5(b), respectively. Based on the projections along the *b* and *c* axes of the *o*-BC<sub>2</sub>N structure, we can confirm that the electron orbits in the A, C, and D regions formed one dimensional chain distributions (1D), respectively, around C<sub>2</sub>, C<sub>6</sub>-C<sub>7</sub> ([100] direction) and B<sub>4</sub>-C<sub>8</sub> ([010] direction) atoms. In the B region, the electron orbits formed two dimensional distributions (2D) rounding B<sub>2</sub>-C<sub>3</sub> atoms along the [100] direction and C<sub>3</sub>-C<sub>4</sub> atoms following the [010] direction.

Combined with the analyses of the energy band structures and electron orbits, we consider that the electrons in the *o*-BC<sub>2</sub>N structure can conduct through the linked electron orbits near the Fermi level along the orientation parallel to the [100] and [010] directions in different layers, which indicate that the new *o*-BC<sub>2</sub>N crystal may possess a fascinating electronic property with 1D and 2D alternate metallicity.

Based on our semi-empirical microscopic hardness theoretical model,<sup>32-35</sup> the Vickers hardness of *o*-BC<sub>2</sub>N can be calculated by the formula as follows:  $H_v(\text{GPa}) = AN_e^{2/3} d^{-2.5} e^{-1.191f_i - 32.2f_m^{0.55}}$ . In the formula, *A* is a content 350, *N<sub>e</sub>* is the electron density, *d* is the bond length of each *x*-*y* chemical bond, *f<sub>i</sub>* is the Phillips ionicity of the chemical bond, which can be calculated by  $f_i = [1 - \exp(-|P_c - P|/P)]^{0.735}$ , and *f<sub>m</sub>* is a factor of metallicity written as  $f_m = 0.026D_F/n_e$ , where *D<sub>F</sub>* is the electron density of states at the Fermi level, and *n<sub>e</sub>* is the total number of the valence electrons in a unit cell.<sup>35</sup> Bond parameters of the *o*-BC<sub>2</sub>N crystal obtained from first-principles calculations are listed in Table III. Here we obtained the pure covalent population

*P<sub>c</sub>* by calculation of the overlap population of the C-C bonds in the supercell with 128 carbon atoms stacking on the basis of the *o*-BC<sub>2</sub>N structure. The calculated Vickers hardness of the *o*-BC<sub>2</sub>N is 41.2 GPa, which indicates that it is a potential superhard material. Combining with the novel electronic property of *o*-BC<sub>2</sub>N, this unique crystallographic characteristic of *o*-BC<sub>2</sub>N is expected to possess some potential applications in electronic devices, such as metal-solid dielectric-vacuum junction, metal-insulator “multiple quantum well” devices, and photonic devices. Meanwhile, the *o*-BC<sub>2</sub>N crystal may be manufactured into a special anvil with conductivity used in high pressure installations.

#### IV. CONCLUSIONS

A distinctive *sp*<sup>2</sup>-*sp*<sup>3</sup> hybridized *o*-BC<sub>2</sub>N structure is predicted. The *o*-BC<sub>2</sub>N is constructed from C layers sandwiched between BN layers and contains *sp*<sup>2</sup>- and *sp*<sup>3</sup>-hybridized B, C, and N atoms. Calculations of elastic constants and phonon frequencies have confirmed the structural stability of *o*-BC<sub>2</sub>N. The electronic band structures, DOS, and PDOS

TABLE III. Chemical bond parameters and Vickers hardness of *o*-BC<sub>2</sub>N.

Bond type	<i>d</i>	Pop	<i>P<sub>c</sub></i>	<i>N<sub>e</sub></i>	<i>f<sub>i</sub></i>	<i>f<sub>m</sub></i> (10 <sup>-3</sup> )	<i>H<sub>v</sub></i>
C1-C2	1.521	0.85	0.87	0.629	0.063	0.627	41.2
C2-C3	1.359	1.43	1.47	1.009	0.071	1.864	
C3-C4	1.558	0.85	0.87	0.586	0.063	1.161	
C4-C5	1.517	0.86	0.87	0.635	0.037	0.611	
C5-C6	1.338	1.45	1.47	1.056	0.042	0.864	
C7-C8	1.544	0.87	0.87	0.602	0	0.928	
C2-N1	1.304	1.21	1.47	1.284	0.298	0.777	
C6-N3	1.519	0.69	0.90	0.700	0.374	0.388	
B2-C2	1.570	0.92	0.90	0.511	0.059	1.153	
B4-C7	1.465	1.30	1.47	0.704	0.213	1.636	
B1-N1	1.545	0.72	0.90	0.497	0.329	0.324	
B1-N2	1.577	0.77	0.80	0.483	0.090	0.204	
B2-N2	1.597	0.65	0.80	0.466	0.313	0.220	
B3-N3	1.611	0.66	0.80	0.454	0.296	0.150	
B3-N4	1.551	0.72	0.80	0.509	0.191	0.145	
B4-N4	1.567	0.72	0.80	0.555	0.191	0.775	

show that the *o*-BC<sub>2</sub>N crystal exhibits a metallic character, and the conducting electrons at the Fermi level are mostly from the *2p* orbitals of *sp*<sup>2</sup>-bonded atoms, with small contributions from the *sp*<sup>3</sup>-bonded B<sub>2</sub> atom. Studies on electron orbits demonstrate that the electrons can transmit along the parallel forming around the C<sub>2</sub>, C<sub>6</sub>, and C<sub>7</sub> atoms along the [100] direction, and around the B<sub>4</sub>, and C<sub>8</sub> atoms along the [010] direction, and on the basal planes consisting of the B<sub>2</sub>-C<sub>3</sub>-C<sub>4</sub> atoms in the *o*-BC<sub>2</sub>N crystal. Consequently, the *o*-BC<sub>2</sub>N possesses a fascinating electronic property characterized by 1D and 2D metallicity, which is distinct from the conventional BC<sub>2</sub>N phases, namely, semimetals, semiconductors, or insulators. The theoretical Vickers hardness of *o*-BC<sub>2</sub>N is approximately 41.2 GPa, which is suggestive of its superhard property. This study of the unique metallic BC<sub>2</sub>N structure offers a new method to explore the ternary B-C-N compounds possessing diverse electrical properties, rendering immense application prospects for these compounds in electronic devices.

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