# Absence of proton tunneling during the hydrogen-bond symmetrization in δ-AlOOH

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 $\delta$ -AlOOH is of significant crystallochemical interest due to a subtle structural transition near 10 GPa from a  $P2_1nm$  to a *Pnnm* structure, the nature and origin of hydrogen disorder, the symmetrization of the O-H···O hydrogen bond and their interplay. We perform a series of density functional theory-based simulations in combination with high-pressure nuclear magnetic resonance (NMR) experiments on  $\delta$ -AlOOH up to 40 GPa with the goal to better characterize the hydrogen potential and therefore the nature of hydrogen disorder. Simulations predict a phase transition in agreement with our NMR experiments at 10 – 11 GPa and hydrogen bond symmetrization at 14.7 GPa. Calculated hydrogen potentials do not show any double-well character and there is no evidence for proton tunneling in our NMR data.

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### I. INTRODUCTION

Hydrogen is an important chemical component in the Earth's mantle, as even a small amount can strongly affect key properties of minerals, such as melting temperature, rheology, electrical conductivity, and atomic diffusion [1–4]. Therefore, over the past 20 years, many hydrous minerals, such as dense hydrous magnesium silicates [5], have been synthesized at high-pressure (*P*) and high-temperature (*T*) conditions and investigated as potential candidates for hydrogen transport to the lower mantle. However, most of these minerals decompose at P < 60 GPa, where phase H breaks down to MgSiO<sub>3</sub> bridgmanite and a fluid component [6–8].

In 2017, AlSiO<sub>3</sub>(OH) was found in diamond inclusions [9] from the mantle transition zone at a depth of 410–660 km. High-*P*, high-*T* experiments revealed that this phase can form from hydrous sediment components at upper mantle conditions (10–12 GPa) [10] and decomposes to  $\delta$ -AlOOH and SiO<sub>2</sub> stishovite at conditions similar to those found at the base of the mantle transition zone (*P* > 20 GPa) [11].  $\delta$ -AlOOH is particularly interesting as it shows a wide stability range including conditions along the geotherm of a subducting slab [12–14] and may therefore be a potential host of hydrogen in Earth's lower mantle.

 $\delta$ -AlOOH crystallizes in a primitive orthorhombic lattice with space group  $P2_1nm$  at ambient conditions [12,15–17]. Its

structure corresponds to distorted rutile, with Al and O atoms located on mirror planes (Fig. 1). AlO<sub>6</sub> octahedra share edges along the *c* axis and these octahedra chains are connected via corners (Fig. 1). There are two distinct oxygen positions (O1 and O2) at the vertices and in the equatorial plane, respectively. Layers of AlO<sub>6</sub> octahedra oriented in different directions are connected with an asymmetric hydrogen bond between the layers.

Single-crystal synchrotron x-ray diffraction (XRD) [15,18] and neutron diffraction measurements [19,20] identified a sub- to supergroup phase transition from  $P2_1nm$  to Pnnm at ~10 GPa where O1 and O2 positions become equivalent. The transition is accompanied by a rotation of the AlO<sub>6</sub> octahedra by ~1°, and in the same P range the hydrogen position is predicted to symmetrize [18,20–23]. In Fourier difference maps calculated from their neutron diffraction data, Sano-Furukawa *et al.* [20] found a bimodal hydrogen distribution between 9.5 and 18 GPa, supporting previous suggestions [18,21,22] that a double-well potential along the diagonal O-O direction may exist, which could give rise to proton tunneling, similar to the state found during the ice-VII to ice-X transition [24–26]. No direct evidence of tunneling has been found to date, however.

We investigate the phase transition, hydrogen bond symmetrization (a central unimodal proton distribution between the two respective oxygen atoms), and the possibility of proton tunneling in  $\delta$ -AlOOH, combining density functional theory (DFT)-based calculations and high- and low-field high-*P* nuclear magnetic resonance (NMR) spectroscopy. With DFT, we perform a stepwise optimization of the host lattice and the hydrogen positions over a wide volume (*V*) range and analyze the potential seen by the hydrogen atoms as well as the geometry of the AlO<sub>6</sub> octahedra. We analyze the signal shift as well as the full width at half maximum (FWHM) of our high-field NMR experiments at *P* of up to 40 GPa (c.f. Supplemental Fig. S1 in the Supplemental Material [27]),



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searching for characteristic features of a phase transition and using low-field NMR data at 5.6 GPa, we investigate indications of proton tunneling [25].

### **II. COMPUTATIONAL DETAILS**

All DFT simulations are performed using the Quantum ESPRESSO package [28,29], where possible GPU-accelerated [30]. We use projector augmented wave atomic files for Al, O, and H based on the PBEsol [31] approximation to exchange-correlation which was previously found to show excellent agreement with experimental data [22]. For Al the 2p electrons and lower and for O the 1s electrons are treated as semirelativistic core states. Convergence tests with a threshold of  $10^{-5}$  Ry/atom lead to a Monkhorst-Pack k-point grid [32] of  $12 \times 8 \times 8$  for primitive unit cells and a cutoff energy for the plane-wave expansion of 140 Ry. We use the FINDSYM code [33] for symmetry analysis and VESTA 3 [34] for visualization of structures.

## **III. EXPERIMENTAL DETAILS**

High-*P* cells with pairs of 250- $\mu$ m culet diamond anvils are used, and the preparation of the NMR experiments closely follows the procedure outlined in our previous work [35]; 3.2 mm excitation coils are formed from single turn cover inductors made from 50  $\mu$ m of copper-coated teflon foil. The diamond anvil is coated with 1  $\mu$ m of copper using physical vapor deposition, and subsequently Lenz lens resonators are cut out of this layer using focused ion beam milling.

A 20- $\mu$ m<sup>3</sup> crystal of  $\delta$ -AlOOH (synthesis is described in Ref. [36]) is placed in the sample chamber and the DAC is filled with neon as a *P*-transmitting medium. Both excitation coils are mounted central to the diamond anvils and connected in a Helmholtz coil arrangement after closing the cell. For the high-field measurements, 1045 mT with a corresponding <sup>1</sup>H resonance frequency of  $\sim$ 45 MHz in an electromagnet is used. Additional homonuclear <sup>1</sup>H - <sup>1</sup>H decoupling experiments are conducted using a Lee-Goldburg [37] saturation pulse of 25 W prior to the spin excitation to obtain high-resolution <sup>1</sup>H-NMR spectra. An additional DAC prepared in a similar manner is filled with distilled H<sub>2</sub>O and used as a resonance shift reference. Low-field measurements are performed at 125 mT and a resonance frequency of  $\sim 5$  MHz, using the same electromagnet. Resulting spectra are analyzed by line form matching to the experimental signal [38]. Pressure is calculated from the Raman signal of the diamonds [39,40] and the AlOOH volume using the equation-of-state (EOS) from Simonova *et al.* [36].

# IV. STRUCTURAL OPTIMIZATION AND HYDROGEN POTENTIAL

In the computations, we start structural optimization with reported experimental low-*P* structures from Komatsu *et al.* [15] and Sano-Furukawa *et al.* [20] and optimize the hydrogen position for *V* between 57 and 47 Å<sup>3</sup>. We sample and optimize the hydrogen positions along the diagonal oxygen-oxygen direction first, relax the coordinates of all atoms, perform a second sampling and optimization of the hydrogen positions,



FIG. 1. Crystal structure of  $P2_1nm \delta$ -AlOOH. Aluminum is shown in blue, oxygen in red, and hydrogen in white. Offset of the hydrogen positions from the center is exaggerated for illustration.

relax the cell parameters, and perform a final sampling and optimization of the hydrogen positions. Both initial structures [15,20] converge to the same coordinates within the first steps of the relaxation.

We sample configurations by displacing the hydrogen atoms along the diagonal oxygen-oxygen direction in the *a-b* plane by starting with both hydrogen atoms being close to the respective oxygen atom with the smaller *b* coordinate (normal configuration) and from the configuration shown in Fig. 1 (inverse configuration, *c.f.* Supplemental Fig. S2 of the Supplemental Material [27]). From the energy obtained in each sampling step, we construct the potential seen by the hydrogen atom, similar to our previous work on the ice-VII to ice-X transition [26] via spline interpolation at each sampled V for both configurations. The final positions of the hydrogen atoms are obtained as the minima of the respective spline interpolation (Fig. 2).

We find distinctly different potential symmetries for normal and inverse sampling with respect to the center of the diagonal oxygen-oxygen distance: Normal sampling results in a symmetric potential; inverse sampling results in an asymmetric potential (~0.2 eV lower in energy). Under compression the potentials become narrower in both sampling types, and for inverse sampling asymmetry decreases. At  $V \leq 52.4 \pm 0.1$  Å<sup>3</sup> the potential obtained by inverse sampling becomes symmetric and the energy difference between the two potentials approaches zero, with the inverse sampling remaining slightly lower in energy over the full V range.

Contradicting prior suggestions [21,22], both potentials do not show any double-well character, even though calculating the potential seen by the hydrogen atom from Kohn-Sham DFT should strongly overestimate the potential well without further consideration of the quantum nature of the hydrogen atoms [22,26,41].

After each optimization step, we analyze the space group of the resulting cell and track the rotation of  $AlO_6$  octahedra



FIG. 2. Comparison of normal (dotted) and inverse (solid) sampling (Supplemental Fig. S2 in the Supplemental Material [27]) of the potential seen by the hydrogen atom. At large volumes (low *P*), normal sampling leads to a symmetric and inverse sampling to an asymmetric single-well potential with a significantly lower energy ( $\sim 0.2 \text{ eV}$ ). With increasing compression, this difference vanishes as the two oxygen positions become equivalent. By construction, the potentials are equivalent at the central position (indicated by the vertical dashed line).

using angles  $\omega$  and  $\omega'$  as defined by Sano-Furukawa *et al.* [20] as a function of compression (Fig. 3).

We find an increase in the angle of ~0.6° when compressing from 57 Å<sup>3</sup> to ~53 Å<sup>3</sup>, followed by a decrease at higher compression for both  $\omega$  and  $\omega'$ . The angles are in general ~0.5° larger than the experimental values by Sano-Furukawa *et al.* [20] which show large scatter; the difference between  $\omega$  and  $\omega'$  and the *P* dependence they report are in very good agreement with our calculations.

## V. NMR SPECTROSCOPY

NMR experiments employing Lee-Goldburg decoupling pulses lead to line widths of ~1.5 ppm [shown for 8 and 11 GPa in Fig. 4(a)] which permits the analysis of chemical shifts with ~10 ppm [(Fig. 4(c)]. For the chemical shift, we find an abrupt change at  $P = 9.9 \pm 0.6$  GPa from ~5 to >12 ppm, indicating a structural change in the chemical environment of the hydrogen atoms as expected for a phase transition. We find a single proton signal over the full pressure range indicating a unique, geometrically well-defined proton position, in agreement with an earlier NMR study at ambient conditions by Xue and Kanzaki [42].

Lee-Goldburg decoupling, while retaining information about the isotropic chemical shift interactions and strongly focusing the line width, leads to a cancellation of the nonsecular parts of the total spin Hamiltonian and thus to a loss of information about spin interactions due to chemical shift anisotropy, direct homo- and heteronuclear dipole-dipole, as



FIG. 3. Angle analysis for the fully relaxed cell (filled symbols) and the experimental values by Sano-Furukawa *et al.* [20] (open symbols, SF+2018) following their definition (inset).

well as first-order quadrupolar interactions. Therefore, we additionally analyze the line shape of <sup>1</sup>H-NMR solid-echoes (Supplemental Fig. S1 in the Supplemental Material [27]) and find a minimum in the line width at  $P = 12.7 \pm 0.4$  GPa (Fig. 4(d)), indicating a change in proton mobility.

Proton tunneling should lead to a zero-field splitting accompanied by detectable tunneling side bands as it introduces an exchange between allowed magnetic transitions with  $\Delta m = 1$  and usually forbidden transitions [43]. In work on the ice-VII to ice-X transition in high-P H<sub>2</sub>O [25], we showed that such tunneling side bands can be resolved at high P using low-field NMR, and that NMR is sensitive to changes in the tunneling rate during compression. We therefore employ the low-field setup at P = 5.6 GPa (Fig. 4(b)) and find no indication for tunneling sidebands in  $\delta$ -AlOOH, in agreement with the P evolution of the calculated potentials, showing no double-well character over the respective compression range.

### VI. EQUATION-OF-STATE

In order to convert *V* of the simulation cells to *P*, we use the optimized structures from the inverse configuration and fit a third-order Birch-Murnaghan (BM3) EOS to total energy. We use the *V* at which the potential symmetrizes (52.4 Å<sup>3</sup>) to split the E - V results in two sets and calculate an EOS for (i) the full *V* range, (ii) 57  $\ge V \ge$  53.5 Å<sup>3</sup>, and (iii) 52  $\ge$  $V \ge$  47 Å<sup>3</sup> (Fig. 5(a)). We calculate the intersection of the



FIG. 4. (a) Lee-Goldburg <sup>1</sup>H-NMR spectra at 8 and 11 GPa at a field of 1045 mT with H<sub>2</sub>O as a chemical shift reference. (b) NMR spectrum of  $\delta$ -AlOOH at 5.6 GPa (shifted to zero) at a field of 125 mT. (c) Shift of the resonance as a function of pressure in the Lee-Goldburg spectra. The shift increases abruptly at  $P = 9.9 \pm 0.6$  GPa from ~5 ppm to >12 ppm. (d) FWHM of the resonance peak as a function of *P*. A minimum in line width occurs at  $P = 12.7 \pm 0.4$  GPa (~52.7Å<sup>3</sup>). The dotted black line is a third-order polynomial fit to the data between 5 and 20 GPa. In panels (c) and (d), the blue vertical lines show the minima of the polynomial fit to the line width data and the green line indicates the jump in the resonance shift with the respective errors indicated by the dotted lines.

enthalpy curves for (ii) and (iii) and estimate the transition P as  $11.3 \pm 0.6$  GPa [horizontal line in Fig. 5(a)], where the error is estimated from the shift when the point closest to the transition in the potential is in-/excluded from the respective V ranges. The phase transition is of second order as we do not find a V collapse.

The calculated EOS parameters (low *P*:  $V_0 = 56.0$  Å<sup>3</sup>,  $K_0 = 183$  GPa,  $K'_0 = 3.7$ ; high *P*:  $V_0 = 55.5$  Å<sup>3</sup>,  $K_0 = 224$  GPa,  $K'_0 = 4.0$ ) are in good agreement with partial EOS parameters by Simonova *et al.* [36] (low *P*:  $V_0 = 56.51(8)$  Å<sup>3</sup>,  $K_0 = 142(5)$  GPa; high-*P* part  $V_0 = 55.56(8)$  Å<sup>3</sup>,  $K_0 = 216.0(5)$  GPa, with  $K'_0 = 4.0$  in both cases) and data from



FIG. 5. (a) Third order Birch-Murnaghan (BM3) EOS fit to the total energy of the optimized structures from the inverse configuration sampling. The full *V* range is fitted by the black curve, V > 53.5 Å<sup>3</sup> by the blue curve, and V < 52 Å<sup>3</sup> by the teal curve. The vertical line shows the estimated transition from the potential analysis, and the horizontal line is the intersection of the enthalpy calculated from the two partial BM3 fits. The inset shows a zoom of the area where low- and high-*P* partial EOS intersect. (b) Comparison of different V - P data and EOS from our calculations with literature data by Kuribayashi *et al.* [18], Simonova *et al.* [36], and Suzuki [44].

TABLE I. Calculated values of the transition pressure from the low- <i>P</i> P2 <sub>1</sub> nm to the high- <i>P</i> Pnnm structure of AlOOH. For the line width
analysis, errors are estimated from the P error and for the chemical shift from the P resolution of the measurement. For the simulations, errors
are estimated from the P resolution of the sampling and for the EOS calculation as the difference between the intersection P of the low-P and
the high-P sub-EOS (Fig. 5). The last three columns show literature data from Simonova et al. [36] (S+ 2020), Sano-Furukawa et al. [20]
(SF+ 2018), and Pillai <i>et al.</i> [21] (P+ 2018).

Criterion	NMR Shift	EOS	< AlOAl > angles	NMR line width	Potential	S+ 2020	SF+ 2018	P+ 2018
P (GPa)	$9.9\pm0.6$	$11.3\pm0.6$	$13.1\pm0.7$	$12.4\pm0.3$	$14.7\pm0.4$	>10 (structural)	9/18.1	8/15

Suzuki [44] and Kuribayashi *et al.* [18] at V < 55 Å<sup>3</sup> (Fig. 5(b)). At larger V, our EOS slightly underestimates P compared with the experimental EOS, which is most likely caused by thermal effects not being included in the calculations.

Using the respective low-*P* and high-*P* EOS parameters, we convert all *V* dependencies to *P* dependencies and find potential symmetrization at  $P = 14.7 \pm 0.4$  GPa and the maxima in  $\omega$  and  $\omega'$  at  $P = 13.1 \pm 0.7$  GPa (Table I).

#### VII. DISCUSSION AND CONCLUSION

We have analyzed different properties of  $\delta$ -AlOOH that can be directly linked to the phase transition (chemical shift, EOS) and hydrogen bond symmetrization (potential symmetry), but also properties that should record an influence of both processes (angles  $\omega$  and  $\omega'$  as well as the NMR line width). We find features in the same three *P* regions: (i) The phase transition characterized by the change in chemical shift at 9.9  $\pm$  0.6 GPa and the change in the slope of the *E* - *V* curve (and therefore splitting of the EOS) at 11.3  $\pm$  0.4 GPa; (ii) a maximum in the angles  $\omega$  and  $\omega'$  at 13.1  $\pm$  0.7 GPa and a minimum in the line width at 12.4  $\pm$  0.3 GPa; (iii) the symmetrization of the potential at 14.7  $\pm$  0.4 GPa.

Comparing our results with recently published experiments [20,36] and computations [21,22], we find that we match the phase transition measured via neutron diffraction by Sano-Furukawa *et al.* [20] and XRD by Simonova *et al.* [36] at ~10 GPa in the NMR experiments and the calculations. Furthermore, we reproduce the hydrogen bond symmetrization estimate from the calculation of elastic constants by Cortona [22] and Pillai *et al.* [21] at ~15 GPa in our potential analysis. For angles  $\omega$  and  $\omega'$ , we find the same increase in slope as Sano-Furukawa *et al.* [20], with a maximum at 13.1  $\pm$  0.7 GPa. If we take a closer look at the data by Sano-Furukawa *et al.* [20], values for the angles  $\omega$  and  $\omega'$  at the highest three

- D. Bercovici and S.-i. Karato, Nature (London) 425, 39 (2003).
- [2] T. Inoue, Phys. Earth Planet. Inter. 85, 237 (1994).
- [3] S.-I. Karato, M. S. Paterson, and J. D. FitzGerald, J. Geophys. Res.: Solid Earth 91, 8151 (1986).
- [4] T. Yoshino, T. Matsuzaki, S. Yamashita, and T. Katsura, Nature (London) 443, 973 (2006).
- [5] A. B. Thompson, Nature (London) 358, 295 (1992).
- [6] E. Ohtani, Y. Amaike, S. Kamada, T. Sakamaki, and N. Hirao, Geophys. Res. Lett. 41, 8283 (2014).

*P* points ( $P \ge 13$  GPa) appear to decrease linearly with *P*, in agreement with our computational results (Fig. 3). The large scatter and limited *P* range in the experimental data impede a more detailed analysis, comparison, and discussion, however.

The absence of a double well in our mapping of the hydrogen potential at any V is supported by low-field NMR measurements which do not show any indication of tunneling side bands. Therefore, we conclude that, contrary to the ice-VII to ice-X transition [25,26], there is no tunneling-induced proton disorder in  $\delta$ -AlOOH. The only observation directly linked to a double-well potential are the Fourier difference maps by Sano-Furukawa et al. [20] that describe an asymmetric proton distribution at P > 9.5 GPa, followed by a bimodal distribution to  $P \lesssim 15$  GPa and a symmetric unimodal distribution at P = 18 GPa. According to our results and data, the intermediate (bimodal) state is not characterized by proton disorder. Rather, it reflects order with weak asymmetry that gradually decreases. Therefore, we suggest the following interpretation of the neutron data: As O1 and O2 become symmetrically equivalent during the structural transition from  $P2_1nm$  to Pnnm at ~10 GPa, asymmetry can no longer be associated to an  $O1-H\cdots O2$  bond, but an averaged picture emerges where protons are closer to former O1 and O2 atoms, which may be visible in the Fourier difference maps and lead to the bimodal distribution if projected onto a *Pnnm* unit cell.

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- [7] J. Tsuchiya and K. Umemoto, Geophys. Res. Lett. 46, 7333 (2019).
- [8] M. Nishi, J. Tsuchiya, T. Arimoto, S. Kakizawa, T. Kunimoto, Y. Tange, Y. Higo, and T. Irifune, Phys. Chem. Miner. 45, 995 (2018).
- [9] R. Wirth, C. Vollmer, F. Brenker, S. Matsyuk, and F. Kaminsky, Earth Planet. Sci. Lett. 259, 384 (2007).
- [10] S. Ono, J. Geophys. Res.: Solid Earth 103, 18253 (1998).
- [11] A. Sano, E. Ohtani, T. Kubo, and K.-i. Funakoshi, J. Phys. Chem. Solids 65, 1547 (2004).

- [12] A. Sano, E. Ohtani, T. Kondo, N. Hirao, T. Sakai, N. Sata, Y. Ohishi, and T. Kikegawa, Geophys. Res. Lett. 35, L03303 (2008).
- [13] Y. Duan, N. Sun, S. Wang, X. Li, X. Guo, H. Ni, V. B. Prakapenka, and Z. Mao, Earth Planet. Sci. Lett. 494, 92 (2018).
- [14] X. Su, C. Zhao, C. Lv, Y. Zhuang, N. Salke, L. Xu, H. Tang, H. Gou, X. Yu, Q. Sun, and J. Liu, Geosci. Front. 12, 937 (2021).
- [15] K. Komatsu, T. Kuribayashi, A. Sano, E. Ohtani, and Y. Kudoh, Acta Crystallogr., Sect. E: Struct. Rep. Online 62, i216 (2006).
- [16] E. Ohtani, K. Litasov, A. Suzuki, and T. Kondo, Geophys. Res. Lett. 28, 3991 (2001).
- [17] A. Suzuki, E. Ohtani, and T. Kamada, Phys. Chem. Miner. 27, 689 (2000).
- [18] T. Kuribayashi, A. Sano-Furukawa, and T. Nagase, Phys. Chem. Miner. 41, 303 (2014).
- [19] A. Sano-Furukawa, K. Komatsu, C. B. Vanpeteghem, and E. Ohtani, Am. Mineral. 93, 1558 (2008).
- [20] A. Sano-Furukawa, T. Hattori, K. Komatsu, H. Kagi, T. Nagai, J. J. Molaison, A. M. dos Santos, and C. A. Tulk, Sci. Rep. 8, 15520 (2018).
- [21] S. B. Pillai, P. K. Jha, A. Padmalal, D. Maurya, and L. Chamyal, J. Appl. Phys. **123**, 115901 (2018).
- [22] P. Cortona, J. Phys.: Condens. Matter 29, 325505 (2017).
- [23] J. Tsuchiya, T. Tsuchiya, S. Tsuneyuki, and T. Yamanaka, Geophys. Res. Lett. 29, 15 (2002).
- [24] L. Lin, J. A. Morrone, and R. Car, J. Stat. Phys. 145, 365 (2011).
- [25] T. Meier, S. Petitgirard, S. Khandarkhaeva, and L. Dubrovinsky, Nat. Commun. 9, 2766 (2018).
- [26] F. Trybel, M. Cosacchi, T. Meier, V. M. Axt, and G. Steinle-Neumann, Phys. Rev. B 102, 184310 (2020).
- [27] See Supplemental Material at http://link.aps.org/supplemental/ 10.1103/PhysRevB.104.104311 for an additional visualization of the structures and <sup>1</sup>H-NMR solid-echoes.
- [28] P. Giannozzi, S. Baroni, N. Bonini, M. Calandra, R. Car, C. Cavazzoni, D. Ceresoli, G. L. Chiarotti, M. Cococcioni, I. Dabo, A. Dal Corso, S. de Gironcoli, S. Fabris, G. Fratesi, R. Gebauer, U. Gerstmann, C. Gougoussis, A. Kokalj, M. Lazzeri, L. Martin-Samos *et al.*, J. Phys.: Condens. Matter **21**, 395502 (2009).

- [29] P. Giannozzi, O. Andreussi, T. Brumme, O. Bunau, M. B. Nardelli, M. Calandra, R. Car, C. Cavazzoni, D. Ceresoli, M. Cococcioni, N. Colonna, I. Carnimeo, A. D. Corso, S. de Gironcoli, P. Delugas, R. A. D. Jr., A. Ferretti, A. Floris, G. Fratesi, G. Fugallo *et al.*, J. Phys.: Condens. Matter **29**, 465901 (2017).
- [30] J. Romero, E. Phillips, G. Ruetsch, M. Fatica, F. Spiga, and P. Giannozzi, in *International Workshop on Performance Modeling, Benchmarking and Simulation of High Performance Computer Systems* (Springer, Cham, 2017), pp. 67–87.
- [31] J. P. Perdew, A. Ruzsinszky, G. I. Csonka, O. A. Vydrov, G. E. Scuseria, L. A. Constantin, X. Zhou, and K. Burke, Phys. Rev. Lett. 100, 136406 (2008).
- [32] H. J. Monkhorst and J. D. Pack, Phys. Rev. B 13, 5188 (1976).
- [33] H. T. Stokes and D. M. Hatch, J. Appl. Crystallogr. 38, 237 (2005).
- [34] K. Momma and F. Izumi, J. Appl. Crystallogr. 44, 1272 (2011).
- [35] T. Meier, F. Trybel, G. Criniti, D. Laniel, S. Khandarkhaeva, E. Koemets, T. Fedotenko, K. Glazyrin, M. Hanfland, M. Bykov *et al.*, Phys. Rev. B **102**, 165109 (2020).
- [36] D. Simonova, E. Bykova, M. Bykov, T. Kawazoe, A. Simonov, N. Dubrovinskaia, and L. Dubrovinsky, Minerals 10, 1055 (2020).
- [37] T. Meier, S. Khandarkhaeva, J. Jacobs, N. Dubrovinskaia, and L. Dubrovinsky, Appl. Phys. Lett. 115, 131903 (2019).
- [38] T. Meier, D. Laniel, M. Pena-Alvarez, F. Trybel, S. Khandarkhaeva, A. Krupp, J. Jacobs, N. Dubrovinskaia, and L. Dubrovinsky, Nat. Commun. 11, 6334 (2020).
- [39] Y. Akahama and H. Kawamura, J. Appl. Phys. 96, 3748 (2004).
- [40] Y. Akahama and H. Kawamura, J. Appl. Phys. 100, 043516 (2006).
- [41] C. Drechsel-Grau and D. Marx, Phys. Rev. Lett. 112, 148302 (2014).
- [42] X. Xue and M. Kanzaki, J. Phys. Chem. B 111, 13156 (2007).
- [43] S. Clough, A. Horsewill, P.J. McDonald, and F.O. Zelaya, Phys. Rev. Lett. 55, 1794 (1985).
- [44] A. Suzuki, Mineral. Mag. 73, 479 (2009).