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1. Introduction

Recently, as an ideal semiconductor photocatalyst, bismuth vanadate (BiVO₄) has drawn great attention because of its promising photocatalytic properties.^{1–3} BiVO₄ exhibits a rich structural polymorphism, and three main crystalline phases have been observed, namely, the tetragonal zircon structure (tz-BiVO₄, *I*4₁/*amd*), monoclinic fergusonite structure (mf-BiVO₄, *I*2/*a*) and the tetragonal scheelite structure (ts-BiVO₄, *I*4₁/*a*).^{4,5} Among these structures, mf-BiVO₄ shows excellent photocatalytic activity, while tz-BiVO₄ and ts-BiVO₄ exhibit relatively poor photocatalytic activity.^{6,7} Thus, mf-BiVO₄ is not considered as an excellent phosphor host compared to tz-BiVO₄.⁸

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BiVO₄ has attracted much attention in recent years due to its active photocatalytic and microwave dielectric properties. BiVO₄ exhibits a rich structural polymorphism, and its properties strongly depend on the crystalline phase. Therefore, it is of great importance to achieve an easy control of its crystalline phase. In the present work, phase stability and vibrational properties of fergusonite- and zircon-type BiVO₄ are investigated up to 41.6 GPa by *in situ* synchrotron X-ray diffraction (XRD), Raman spectroscopy, and first principles calculation. Upon compression, although having different initial structures, both types of BiVO₄ consecutively transform to scheelite- and β -fergusonite structures. For the first time reported for BiVO₄, the β -fergusonite structure is determined using first principles computational techniques and from refinement of the XRD data. Along the way, one new phase of BiVO₄ is theoretically predicted at higher pressures. Moreover, both the fergusonite-to-scheelite and scheelite-to- β -fergusonite transitions are reversible, while the zircon-to-scheelite transition is irreversible. A large volume collapse is observed associated with each phase transition, and the equations of state for different types of BiVO₄ have been determined. These results provide new insights into the relationship between different structural types in the AVO₄ family.

Therefore, the properties of $BiVO_4$ depend strongly on the crystalline structure. For practical applications, it is of great importance to achieve an easy control of the crystal phase of $BiVO_4$.

Generally, tz-BiVO₄ can be prepared in aqueous media by a low-temperature process. Tz-BiVO₄ can transform to mf-BiVO₄ after heat treatment at 600-770 K by an irreversible transition.9 Besides temperature, pressure has also been extensively used to discover and access new structures and/or novel properties of materials.^{10,11} The structural stability and phase transition of orthovanadate compounds have been explored under high pressure. Most orthovanadates exist in zircon or scheelite structures under ambient conditions.12-19 For most zircontype orthovanadate compounds, such as HoVO₄, TbVO₄ and LuVO₄, a pressure-induced irreversible transition to a scheelite structure is known, while several zircon structures with a larger cation, such as NdVO₄ and PrVO₄, undergo a zircon-to-monazite transition.^{14-16,19} In contrast, scheelite-type orthovanadates, such as GdVO₄ and ScVO₄, have been observed to transform into the fergusonite structure.12,13

Different from most orthovanadates, $BiVO_4$ adopts a fergusonite structure rather than a zircon or scheelite structure under ambient conditions. Therefore, it may exhibit different behaviors under pressure. It has been previously reported that mf-BiVO₄ would transform to ts-BiVO₄ above 1.7 GPa.²⁰⁻²² However, its pressure range is too low. Recently, it was reported that both mf-BiVO₄ and tz-BiVO₄ can transform to ts-BiVO₄ and



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then a new structural change was observed above 15.7 GPa.¹¹ But this new structure cannot be confirmed because of the lack of high-pressure (HP) X-ray diffraction (XRD) data. In another recent study, phase transitions of fergusonite-scheelite- $P2_1/n$ were reported from in situ synchrotron XRD analysis for fergusonite-type $Eu_{0.1}Bi_{0.9}VO_4$.²³ Therefore, the $P2_1/n$ structure may be speculated to be the second high pressure phase for mf-BiVO₄. However, it is well known that doping can induce lattice distortions and reduce the stability of the structure, resulting in a significant effect on the phase transition. With increasing pressure, for example, Y₂O₃ showed a cubic-hexagonal transition, while Y2O3:Eu exhibited a cubic-monoclinic-hexagonal structure sequence.²⁴ Different phase transition sequences have also been observed between other pure and doped samples, such as BiFeO3 and BiFeO3:Ba, SnO2 and SnO2:Fe.25-27 Particularly for mf-BiVO₄, it has been reported that lanthanide doping can trigger phase transformation from mf-BiVO₄ to tetragonal tz-BiVO₄.^{7,28-30} More interestingly, after decompression, one mixed phase has been reported for mf-BiVO₄, which has not been observed in mf-Eu_{0.1}Bi_{0.9}VO₄.^{11,23} Whether this is associated with a phase transition or decomposition could not be confirmed. Furthermore, the high-pressure behavior of zircontype BiVO₄ needs further confirmation, especially using in situ synchrotron XRD. Thus, there are uncharted structural changes for BiVO₄, so it is necessary to extend the previous studies to fully understand the structural stability and polymorphs of BiVO₄ under high pressure, especially to precisely determine the crystal structures using high-pressure XRD.

In view of this aim, in this work, fergusonite- and zircon-type $BiVO_4$ are successfully synthetized using the hydrothermal method. Then, *in situ* high-pressure XRD and Raman experiments are performed at room temperature. Several phase transitions are observed including the one to a new crystalline phase. The structure of the new phase was characterized using *ab initio* computational techniques. These results reveal a microscopic picture of the pressure-induced phase transition nature for the fergusonite- and zircon-type ABO₄ compounds and provide insights into underlying structure-property relationships of $BiVO_4$.

2. Experimental and theoretical methods

2.1 Sample preparation

Pure mf- and tz-BiVO₄ samples were synthesized using the hydrothermal method.^{30,31} The starting materials were $Bi(NO_3)_3 \cdot 5H_2O$ and NH_4VO_3 with a purity of 99.9%. First, 2.43 g of $Bi(NO_3)_3 \cdot 5H_2O$ was dissolved into 25 mL of 2.0 mol L⁻¹ HNO_3 , which was marked as solution-A, while 0.58 g of NH_4VO_3 was dissolved into 10.0 mL of 5.0 mol L⁻¹ NaOH, which was marked as solution-B. Then, solution-B was added dropwise to solution-A under constant magnetic stirring. After stirring for 30 min, some amount of 1 mol L⁻¹ NaOH aqueous solution was added dropwise with stirring to tune the final pH of the growth solution to 5.0. Then, the suspension was stirred for another 10 min and poured into a 50 mL Teflon-lined stainless autoclave. Subsequently, it was sealed at 220 °C for 10 hours, and then allowed to cool to room temperature naturally. Finally, the precipitation was separated by centrifugation, washed with distilled water and absolute ethanol, and dried at 80 °C to obtain the tz-BiVO₄ sample. Part of the tz-BiVO₄ sample was annealed at 600 °C for 10 hours to transform into the mf-BiVO₄ sample. The crystal structures of the two samples were firstly identified by X-ray diffraction using Cu K α radiation (1.5418 Å). The XRD and Raman results confirmed the successful synthesis of mf- and tz-BiVO₄ samples, as shown in Fig. S1 and S2 (ESI†).

2.2 High-pressure measurements

The high-pressure experiment was carried out to the sample in a diamond anvil cell (DAC), with two diamonds of 350 micron size. The pressure was measured by the shift of the R1 photoluminescence line of ruby.^{32,33} The Raman measurements were performed by using a Renishaw InVia Raman microscope with a 532 nm laser. The power of the laser is 5 mW to avoid the laser heating effect. *In situ* XRD experiments were performed at the 16-BMD beamline of the HPCAT facility at the Advanced Photon Source (APS) using monochromatic radiation with $\lambda = 0.4959$ Å. Diffraction images were recorded using a Mar345 image plate detector, 230 mm away from the sample, and were integrated and corrected for distortions using the FIT2D software. Argon (Ar) was used as the pressure transmitting medium (PTM) for high-pressure Raman spectroscopy, while silicon oil was used as the PTM for XRD.

2.3 Computational methods

Pressure-induced phase transitions of BiVO₄ were simulated using the metadynamics method^{34,35} combined with the projector augmented plane-wave (PAW) method³⁶ as implemented in the Vienna Ab Initio Simulation (VASP) program.³⁷ PAW potentials with the Perdew–Burke–Ernzerhof (PBE) functional³⁸ and a 500 eV energy cutoff were used. The Bi, V, and O potentials employ 5d¹⁰6s²6p³, 3s²3p⁶3d³4s², and 2s²2p⁴ as valence states, respectively. Each metastep consisted of a first principles molecular dynamics (MD) simulation within the canonical (NVT) ensemble for a simulation time of 0.4 ps, along with a *k*-spacing of $2\pi \times 0.08 \text{ Å}^{-1}$ for the Brillouin zone (BZ) sampling. To reduce the energy barrier for phase transitions, the supercells were over-pressurized under both hydrostatic and nonhydrostatic conditions. A recent implementation³⁹ was used to load the uniaxial stresses to the simulation cell. Enthalpy calculation for candidate structures was conducted with a fine k-spacing of $2\pi \times 0.03$ Å⁻¹, which yielded excellent convergence for total energy (within 1 meV per atom). Phonon dispersion relations were calculated using the density functional perturbation theory40 and post processed using the PHONOPY code.41

3 Results and discussion

3.1 High-pressure Raman spectra of BiVO₄

The Raman spectra of the prepared $BiVO_4$ samples measured under ambient conditions are presented in Fig. S2 (ESI†).

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Fig. 1 Selected high-pressure Raman spectra of (a) mf-BiVO₄ and (b) tz-BiVO₄ upon compression. The pressures in GPa are labelled for each spectrum. The asterisks denote the new modes that emerged.

The assignments of Raman active modes confirm that the prepared samples are indeed in fergusonite- and zircon-type structures.^{2,9,42} For mf-BiVO₄, upon compression, a fergusoniteto-scheelite transition is observed above 2.1 GPa, as shown in Fig. 1(a).²⁰⁻²² Fergusonite is a weakly distorted version of the scheelite structure, and therefore most Raman modes are retained during this transition.¹¹ After phase transition, two otherwise split lines (323.5 and 365.6 cm⁻¹) in mf-BiVO₄ gradually merge into one peak. Upon subsequent compression to 13.6 GPa, no other change is detected, except for a phonon softening observed at 116.3 cm⁻¹. Above 17.8 GPa, three new modes emerge at 734.6, 860, and 905.8 cm⁻¹, and their intensity gradually increased during the compression, as marked in Fig. 1(a). Simultaneously, the original modes of ts-BiVO₄ become weaker and several modes finally become unresolvable at higher pressure. Moreover, pressure-induced broadening is observed above 13.6 GPa because of the non-hydrostatic condition at high pressure. A similar phase transition was also observed for tz-BiVO₄, as shown in Fig. 1(b). A new phase emerges at 1.3 GPa and coexists with the zircon phase until 5.0 GPa. By comparing the Raman spectra of the pure new phase above 5.0 GPa to those of known structures, we found that the new phase is also ts-BiVO₄. Similar to mf-BiVO₄, upon further increasing the pressure above 15.0 GPa, three new bands appeared, and the intensity of the original peaks became somewhat weak with increasing pressure, implying that another structural change has occurred.

The structural evolution of mf-BiVO₄ and tz-BiVO₄ was further established from the pressure dependence of the mode frequencies. As shown in Fig. 2(a) and (b), two inflection points are clearly observed at 3.2 and 17.8 GPa for mf-BiVO₄, while four pressure regions with different structural characters are observed for tz-BiVO₄. Although the mf- and tz-BiVO₄ phases have different structures, both exhibit similar structural evolution and transform into ts-BiVO₄ at high pressure. However, the range of stability for ts-BiVO₄ is small. Above 15 GPa, the second structure transition occurs in both samples, as confirmed by the appearance of several new modes in the Raman spectra, which agrees well with a previous report.¹¹ However, the previous report could not conclude whether this change corresponds to a phase transition or decomposition.¹¹ Thus, it is necessary to investigate the structural evolution upon decompression, because such structural changes would not be reversible if decomposition has happened.

The Raman spectra of both samples upon decompression are shown in Fig. 3. For comparison, the Raman spectra of the original mf-BiVO₄ and tz-BiVO₄ are also presented. The results show that the two samples exhibit the same structural evolution during decompression. Both high-pressure structures (yet to be determined) obtained by compressing mf-BiVO₄ and tz-BiVO₄ transform back to the same ts-BiVO₄ phase when the pressure is released, at 13.8 and 12.1 GPa, respectively. Upon decompressing further, both samples finally transform back to mf-BiVO₄ below 2 GPa, which can be clearly distinguished from the characteristic splitting modes at 323.5 cm⁻¹ and 365.6 cm⁻¹. The initial structure of tz-BiVO₄ was not observed after pressure release. This result is quite different from the previous report that a mixed phase was obtained after pressure release.¹¹ From the current observation, a reversible phase transition for mf-BiVO₄ during decompression provides a direct explanation for phase transition rather than decomposition. Going further, we need to determine the structures for both high-pressure phases, in particular the newly discovered phase, using synchrotron XRD measurement.

3.2 High-pressure XRD of BiVO₄

The evolutions of the XRD pattern during the compression of two types of BiVO₄ followed by a decompression to ambient

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Fig. 2 Pressure dependence of the Raman mode frequencies of (a) mf-BiVO₄ and (b) tz-BiVO₄ upon compression.



Fig. 3 Selected Raman spectra of (a) mf-BiVO₄ and (b) tz-BiVO₄ upon decompression. For comparison, the Raman spectra of mf-BiVO₄ and tz-BiVO₄ are also shown as a reference.

pressure are shown in Fig. 4. A Rietveld refinement of the XRD patterns was performed using the GSAS software package, as illustrated in Fig. 4(c), while the detailed refinement results are listed in Table 1. For the tz-BiVO₄ sample, the original diffraction peaks disappear, and several new peaks appear at 6.4 GPa, indicating the first phase transition. From the Rietveld refinement, this HP phase is identified as the scheelite structure (ts-BiVO₄), in agreement with the Raman results. The different transition pressures obtained from the Raman results may stem from the effect of the different PTMs. Although the PTM effect is not desirable, it does not influence refinement results significantly. Upon further increasing the pressure to

17.4 GPa, several new diffraction peaks appear at 5.9° , 6.4° and 9.9° (2θ), signaling the second phase transition. Upon further compression, the new peaks gradually grow in intensity, while simultaneously the peaks of ts-BiVO₄ gradually decrease, forming a region of two-phase coexistence.

To identify the post-scheelite structure, first-principles metadynamics simulations were performed. The simulations were carried out at room temperature and at four pressures, *e.g.*, 25, 30, 35, and 40 GPa, starting from a $2 \times 2 \times 2$ ts-BiVO₄ supercell. In the simulations, the structure was overcompressed to allow for a faster reaction. A little to our surprise, the first phase transition was obtained at 30 GPa, where ts-BiVO₄



Fig. 4 XRD patterns of (a) mf-BiVO₄ and (b) tz-BiVO₄ at several selected pressures (in GPa) upon compression to 25.4 and 41.6 GPa, respectively, followed by decompression to ambient pressure (1 atm). (c) Rietveld refinement of the diffraction patterns of tz-BiVO₄. The experimental data are shown as black crosses with refinements shown as red solid lines and the difference between calculated and observed intensities is shown as blue lines. Ticks indicate the positions of Bragg peaks.

transforms directly to a new structure with the C2/c space group, rather than the $P2_1/n$ structure reported for Eu_{0.1}Bi_{0.9}VO₄.²³

Comparing with the known structures in the literature shows that this C_2/c structure is isostructural to β -fergusonite, which has also been observed at high pressure in wolframite ZnWO₄.⁴⁴ To confirm the structure of the post-scheelite phase, we used both P_{2_1}/n and C_2/c (β -fergusonite) structural models to refine the XRD patterns. As can be seen in Fig. S3 (ESI†), the β -fergusonite mode fits the diffraction pattern very well. Moreover, the values of R_p and R_{wp} further suggest that the β -fergusonite mode is more suitable. In addition, the XRD pattern collected at 41.6 GPa can be well refined to a single β -fergusonite structure, as shown in Fig. 4(c), and the unit cell parameters obtained are listed in Table 1.

A similar result was also observed in mf-BiVO₄ up to the maximum pressure of 25.4 GPa. But its spectra are broader at high pressure, which may stem from the non-hydrostatic conditions. Both kinds of BiVO4 share the same structural evolution under high pressure. When the pressure is released in β -fergusonite, the scheelite phase appears again, at pressures below 13.1 GPa. Drastic splitting is observed in several diffraction peaks below 2 GPa, such as 6.1° and $11.1^{\circ} 2\theta$, as marked in the XRD pattern (Fig. 4(a) and (b)). These split peaks correspond to a scheelite-to-fergusonite back transition, which was confirmed by the XRD refinement result collected at 0.1 GPa (Fig. 4(c)). It is very interesting to note that the pressureinduced zircon-to-scheelite transition is not reversible; the scheelite phase would always go back to fergusonite upon decompression. At this point, we conclude that both mf-BiVO₄ and tz-BiVO₄ transform to scheelite and β -fergusonite structures upon compression and go back to only mf-BiVO₄ when the pressure is released.

To understand the pressure-induced polymorphic transitions, the compressibility of each phase was examined using unit cell volume as a function of pressure. Fig. 5 depicts the experimental volume *vs.* pressure data for tz-BiVO₄. These data were fitted using the 3rd-order Birch–Murnaghan (B–M) equation of state (EOS) by fixing the first derivative of bulk modulus at $B_0' = 4$. The ambient pressure bulk modulus B_0 and volume V_0 obtained are also given in Table 1. From the fitting results, the EOS for different types of BiVO₄ have been determined. The bulk modulus $B_0 = 105.2$ GPa and the zero-pressure unit cell

Table 1 Crystal structures and unit cell parameters of BiVO₄ at different pressures obtained from XRD Rietveld refinement. The XRD data were collected during the course of compressing the tz-BiVO₄ sample from ambient pressure and decompressing the final product to ambient pressure

I4 ₁ /amd	$I4_1/a$	C2/c	I2/a
1 atm	6.4 GPa	41.6 GPa	r 1 atm
7.29	5.086	4.858	5.169
7.29	5.086	11.189	5.086
6.45	11.477	4.252	11.497
90	90	89.264	90
(0, 0.75, 0.25)	(0, 0.25, 0.625)	(0.50, 0.402, 0.25)	(0, 0.25, 0.635)
(0, 0.25, 0.375)	(0, 0.25, 0.125)	(0, 0.683, 0.25)	(0, 0.25, 0.155)
(0, 0.072, 0.214)	(0.179, 0.062, 0.261)	(0.675, 0.879, 0.079)	(0.305, 0.637, 0.125)
		(0.372, 0.158, 0.985)	(0.324, 0.344, 0.440)
342.6	306.1	275.9	302.3
101.1	116.3	125.2	111.5^{43}
6.28	7.03	7.80	7.12
6.14	7.58	7.87	6.92
9.42	9.43	10.47	11.72
	$I4_1/amd$ 1 atm 7.29 7.29 6.45 90 (0, 0.75, 0.25) (0, 0.25, 0.375) (0, 0.072, 0.214) 342.6 101.1 6.28 6.14 9.42	$\begin{array}{ccccccc} I4_1/a & I4_1/a \\ 1 \ atm & 6.4 \ GPa \\ 7.29 & 5.086 \\ 7.29 & 5.086 \\ 6.45 & 11.477 \\ 90 & 90 \\ (0, 0.75, 0.25) & (0, 0.25, 0.625) \\ (0, 0.25, 0.375) & (0, 0.25, 0.125) \\ (0, 0.072, 0.214) & (0.179, 0.062, 0.261) \\ 342.6 & 306.1 \\ 101.1 & 116.3 \\ 6.28 & 7.03 \\ 6.14 & 7.58 \\ 9.42 & 9.43 \\ \end{array}$	$\begin{array}{cccccccccccccccccccccccccccccccccccc$



Fig. 5 The pressure dependence of the relative unit cell volume was fitted to the third-order Birch–Murnaghan equation of state.

volume $V_0 = 342.6$ Å³ for tz-BiVO₄ were obtained. The transition from zircon to scheelite involves a volume collapse of approximately 11%, and a larger bulk modulus $B_0 = 116.3$ GPa for ts-BiVO₄. The volume change at the second transition is 8%. Therefore, a high-density β-fergusonite phase is obtained after two HP transitions. The bulk modulus of the β-fergusonite phase is 125.2 GPa, 23.8% larger than that of the initial zircon phase. These values agree well with the theoretical reported values of 103–132 GPa for different polymorphs of BiVO₄.⁴³

3.3 Calculations of high-pressure phases of BiVO₄

To explore other possible high-pressure structures of $BiVO_4$, the enthalpy-pressure relations of the $BiVO_4$ structures were calculated and are shown in Fig. 6. Along with scheelite, zircon, fergusonite, β -fergusonite, and *Cmca* structures, we also considered other structures in related compound groups,



Fig. 6 Enthalpy difference as a function of pressure showing the phase transitions. The scheelite-phase was taken as a zero-energy reference (horizontal black line).

including pseudo-scheelite (P4/nnc), wolframite ($P2_1/c$), $P2_1/n$, M'-fergusonite (P2/a), and CaWO₄-type.^{44–46} From an energetic consideration, tz-BiVO₄ and mf-BiVO₄ will transform to ts-BiVO₄ at around 2 GPa and 5 GPa, respectively. Such transition pressures are very close to the experimental values, *i.e.*, 1.3 GPa and 3.2 GPa. Above 15 GPa, scheelite and β -fergusonite structures have similar enthalpies (indistinguishable in this energy scale), which allows for their coexistence. Below 15 GPa. the β -fergusonite structure would collapse spontaneously to the scheelite structure in the optimization calculation, and therefore its enthalpy in the low-pressure range cannot be obtained. The enthalpy of β -fergusonite becomes lower than scheelite near 30 GPa, indicating a final phase transition in this neighborhood. The predicted theoretical phase transition sequence agrees very well with our experimental observation. Interestingly, β-fergusonite is not the lowest-enthalpy structure in the pressure range of interest. However, the phonon dispersion relations of β -fergusonite show no sign of an imaginary frequency (Fig. S4, ESI[†]), suggesting that this structure is mechanically and thermodynamically stable. The β -fergusonite structure should therefore be considered as a metastable phase stabilized by kinetics. At 30 GPa, metadynamics simulation reveals another phase transition from β -fergusonite to a structure with the *Cmca* space group, which does not belong to any known structures of orthovanadates. The calculated phonon dispersion relation for the Cmca structure shows no imaginary frequency throughout the BZ (Fig. S5, ESI[†]), suggesting that this structure is thermodynamically stable. The no-show of this structure in experiment indicates that it is dynamically blocked with a large activation barrier that requires significant over-compressing. Since the scheelite-to-β-fergusonite transition is just completed at 41.6 GPa, it is reasonable to speculate that the Cmca structure will form at higher pressure. The different value from theoretical calculation may be attributed to the non-hydrostatic condition in practice. We therefore propose the β -fergusonite to *Cmca* transition as a prediction, encouraging future experimental confirmation. In addition, the $P2_1/n$ structure is a distorted version of C2/c with a higher enthalpy. From both experimental fitting and theoretical calculation, the β -fergusonite C2/c structure may be a better candidate because it is dynamically related to the low pressure structure, *i.e.*, it is linked by molecular dynamics through an explainable path.

3.4 Discussion

Fig. 7 shows the observed structural evolution of BiVO₄ under pressure. Although all four structures are composed of BiO₈ dodecahedra and VO₄ tetrahedra, their arrangements are quite different. In zircon and scheelite structures, BiO₈ dodecahedra have two different Bi–O distances, while the four V–O bonds in VO₄ tetrahedron are identical. In the zircon structure, the *c*-axis is formed by alternating VO₄ tetrahedra and BiO₈ dodecahedra, while the *a*-axis is formed by only edge-sharing BiO₈ dodecahedra, resulting in the *c* axis being less compressible than the *a* axis.^{14,16,47} In contrast, in the scheelite structure, the VO₄ tetrahedra are aligned along the *a*-axis, whereas BiO₈ dodecahedra are intercalated with VO₄ tetrahedra along the *c* axis.⁴⁷



Fig. 7 Schematic model for the phase transition of ${\rm BiVO}_4$ upon compression.

Therefore, in the scheelite structure, the *a*-axis is less compressible. During the zircon-to-scheelite transformation, the [110] direction of zircon is converted to the [001] direction of scheelite by a simple shearing mechanism, resulting in a distortion in the VO₄ tetrahedra because of a pressure-induced contraction,⁴⁸ which can be confirmed from the volume collapse of approximately 11% shown in Fig. 5. Consequently, pressure induces a rearrangement of the VO₄ and BiO₈ units during the zircon-to-scheelite phase transition.

The change of the β angle during the scheelite-to- β fergusonite transition indicates that the monoclinic distortion of fergusonite gradually increases under compression. In fergusonite BiVO₄, the V cation is tetrahedrally coordinated to oxygen anions with two different bond lengths, and the Bi cation is coordinated to eight oxygen anions with four different bond lengths, resulting in a lower symmetry. Therefore, during the fergusonite-scheelite transition, the structure symmetry increases, resulting in the merging of several split Raman peaks and XRD peaks, as shown in Fig. 1 and 4. During this phase transition, the tetrahedral coordination of V is retained, which is accompanied by a slight distortion of the VO₄ tetrahedron. It has been pointed out that this phase transition may be driven by the lone-pair Bi³⁺ cation due to the recovery of the distorted dodecahedron and to a lesser extent the VO₄ tetrahedra.⁴ Upon further increasing pressure, ts-BiVO₄ will convert into another fergusonite structure, β -fergusonite, which is known in mineralogy of YNbO₄. Similar to fergusonite, β -fergusonite has also a slight distortion of the scheelite structure with two distinct V-O bonds and four distinct Bi-O bonds. The scheelite to β-fergusonite transition is a first-order reconstructive transformation, resulting in a large volume collapse of about 8%. Consequently, splitting of several initial diffraction peaks in ts-BiVO₄ is observed above 15 GPa during this transition.

It has been proposed that the zircon-scheelite-fergusonite (Z–S–F) transition is a common behavior for most orthovanadate materials under high pressure.^{16,46} However, $BiVO_4$ exhibits different transition sequences, either fergusonite to scheelite

to β -fergusonite, or zircon to scheelite to β -fergusonite. Both starting structures would reach the same β -fergusonite structure under sufficient compression. As discussed above, pressure causes a rearrangement of the VO₄ and BiO₈ units during zircon-to-scheelite transition, resulting in a larger activation barrier. In contrast, the fergusonite-to-scheelite transition is associated with a smaller barrier since only a slight structural distortion is involved. This thermodynamic preference determines that the scheelite structure would transform back to fergusonite rather than the zircon structure during the pressure release. Moreover, from the theoretical prediction (Fig. 6), a new *Cmca* phase is found to be energetically more favorable for BiVO₄ even at a higher pressure. Thus, it is of great interest to examine the possible new polymorphs of BiVO₄ in an extended pressure region in the future.

4. Conclusions

Fergusonite- and zircon-type BiVO₄ materials are successfully synthetized using the hydrothermal method. The high-pressure structural stabilities of both polymorphs have been investigated at room temperature by using in situ high-pressure XRD, Raman spectroscopy, and ab initio calculations. The Raman and XRD results revealed two consecutive reversible transitions for fergusonite BiVO₄, namely fergusonite to scheelite and scheelite to a new HP phase at 3.2 GPa and 15.0 GPa, respectively. For the zircon-type BiVO₄, an irreversible zircon-to-scheelite transition starts at 1.3 GPa and completes at 5.0 GPa. The second transition to the same new HP phase also starts above 15.0 GPa and completes around 41.6 GPa. The new HP phase was successfully identified as β-fergusonite by using metadynamics simulations and refinement of the XRD data. Theoretical enthalpy calculations confirmed the experimental transition sequence and predicted a new high-pressure phase at higher pressures. The high-pressure structure of zircon-type BiVO₄ transformed back to the fergusonite structure upon decompression, indicating that the fergusonite structure is the most stable phase among all known polymorphs of BiVO4. The pressure dependence of the unit-cell parameters has been extracted, and the equations of state of different phases were also determined. The high-pressure behavior of BiVO₄ is different from other orthovanadates, which contributes to a more in-depth understanding of the pressure behavior of AVO₄ compounds.

Conflicts of interest

There are no conflicts to declare.

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